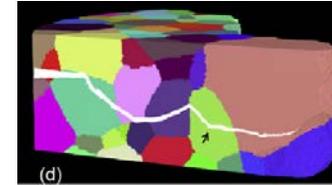
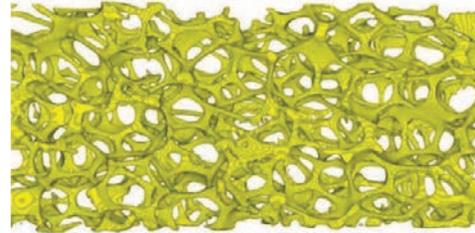


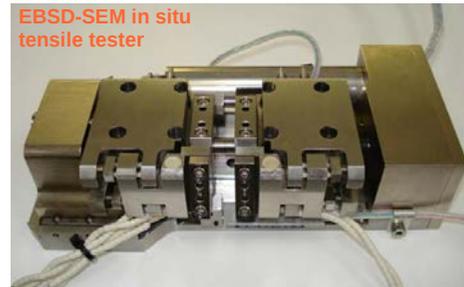




# Context



(d)  
Herbig et al., Acta Mat. 2011



## Damage investigation at MATEIS-METAL

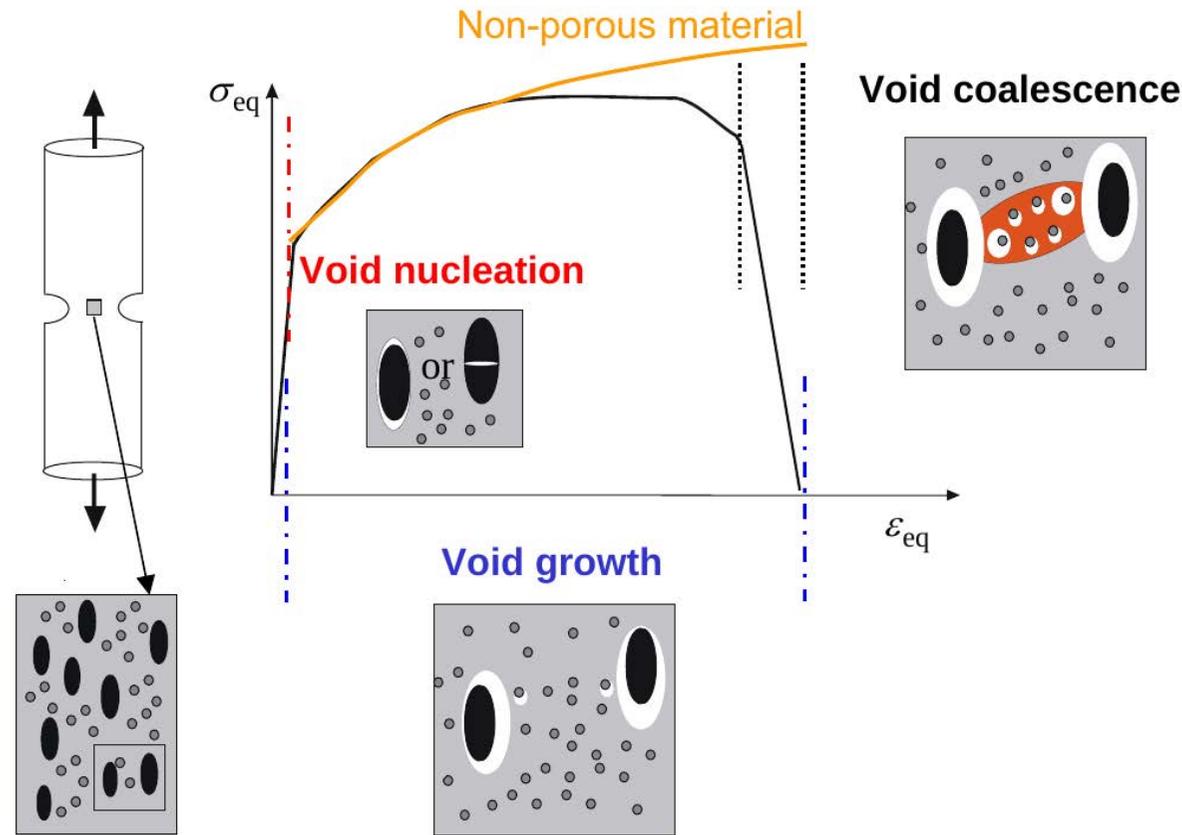
- Structure, microstructure  $\implies$  **damage** development.
- 2D and 3D **in-situ**, **non destructive**, experiments (tension, compression, torsion, indentation, fatigue, ...)







# Ductile damage



Courtesy of Prof. T. Pardoen, UCL

- nucleation, growth and coalescence of cavities
- lots of models (especially for growth)....,
- little experimental comparison (before ~ 2000).

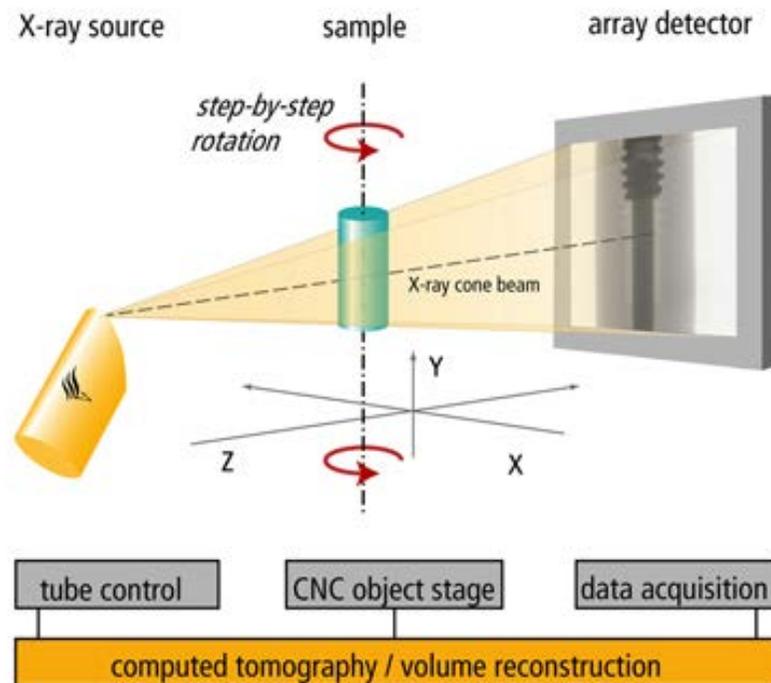




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# X-ray computed tomography

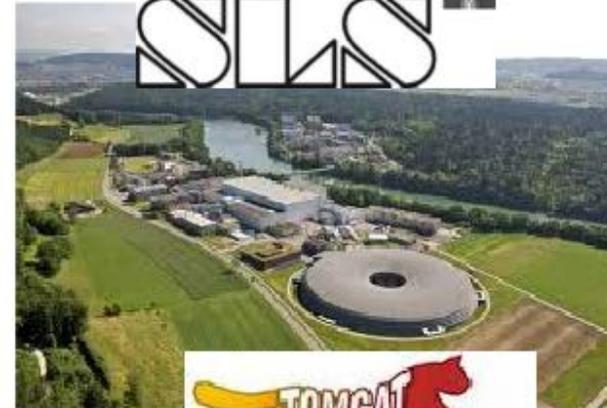


(from Phoenix X ray)

- Lab or synchrotron X-rays.
- Diverging or parallel beam.
- Scan time: 0.05s to hours.
- Resolution: 25nm to centimeters.
- Absorption or phase contrast.
- Non destructive: in-situ/in-operando testing.

Figure: Lab tomography.

# Synchrotron tomographs: high resolution / high speed



APS  
BNL  
Berkeley



ID15 fast acquisition  
ID22 Very high resolution

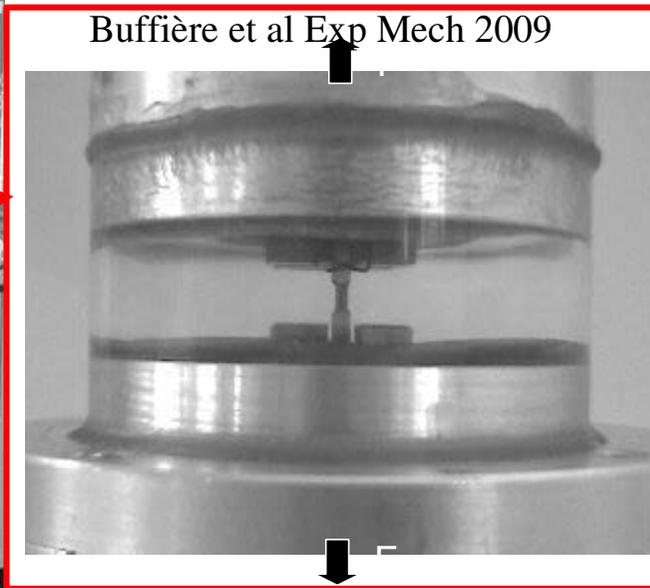
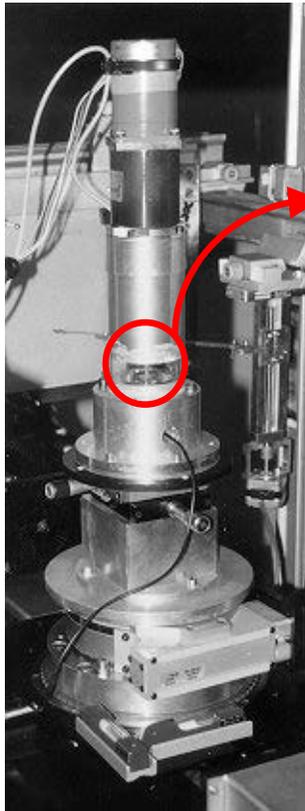


# In situ tensile rig

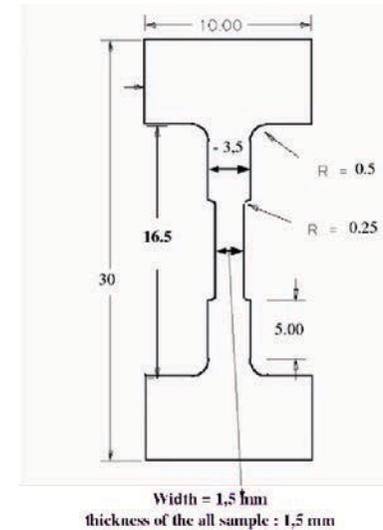
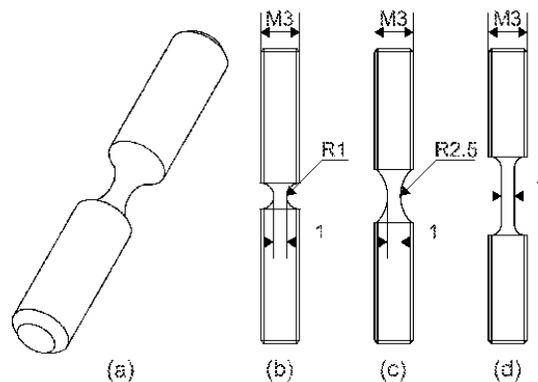
## Tension, compression

Buffière et al. Acta Mater 1998

Buffière et al Exp Mech 2009



- Stepping motor
- Reductor
- F and U recorded
- $10^{-5}$  – 1 mm/s
- Force sensors : 50 – 5000 N
- Grips adapted for different geometries



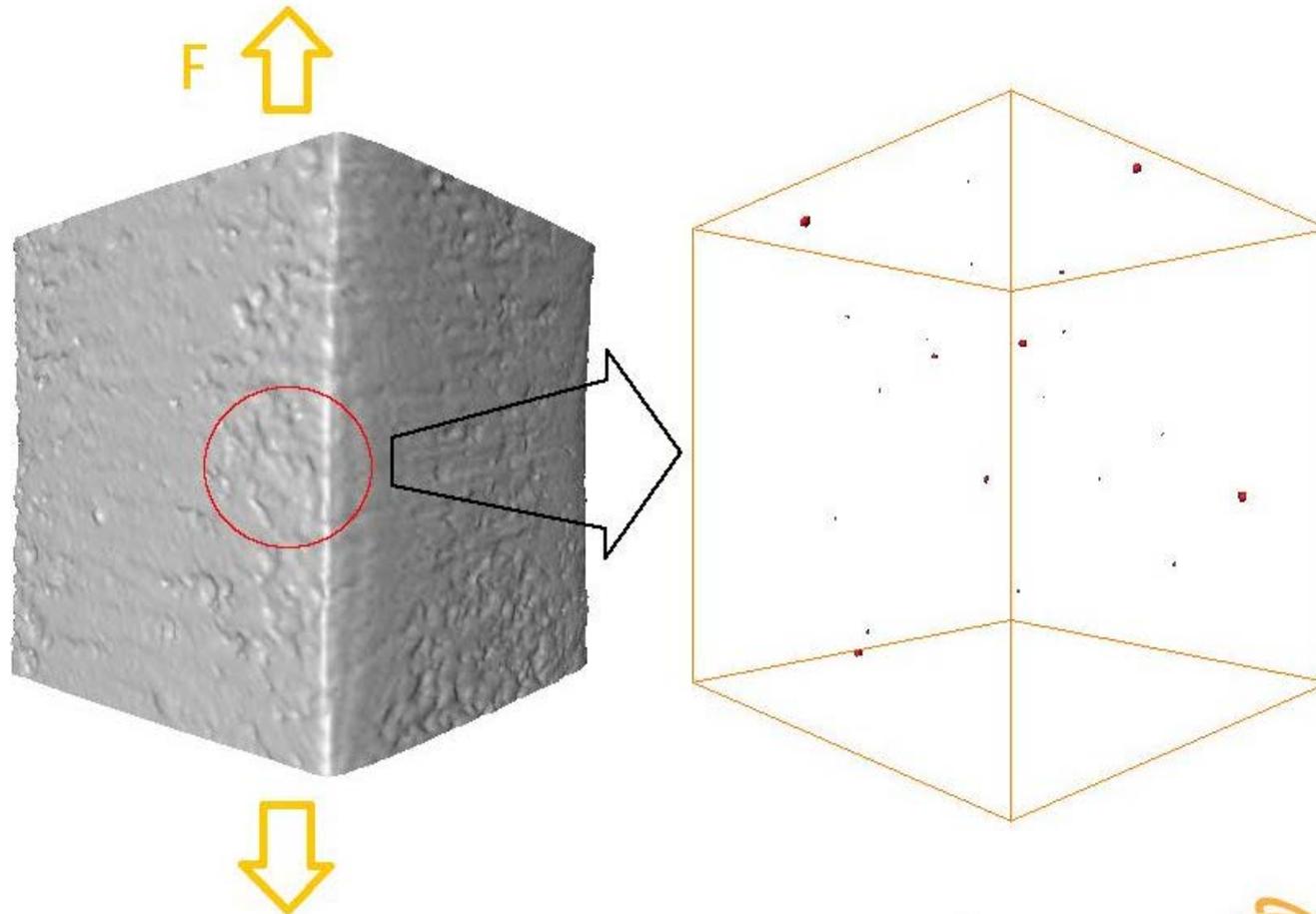
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# Typical result

1 mm sample diameter, ~ 1 micron voxel size, 300 microns central box for analysis

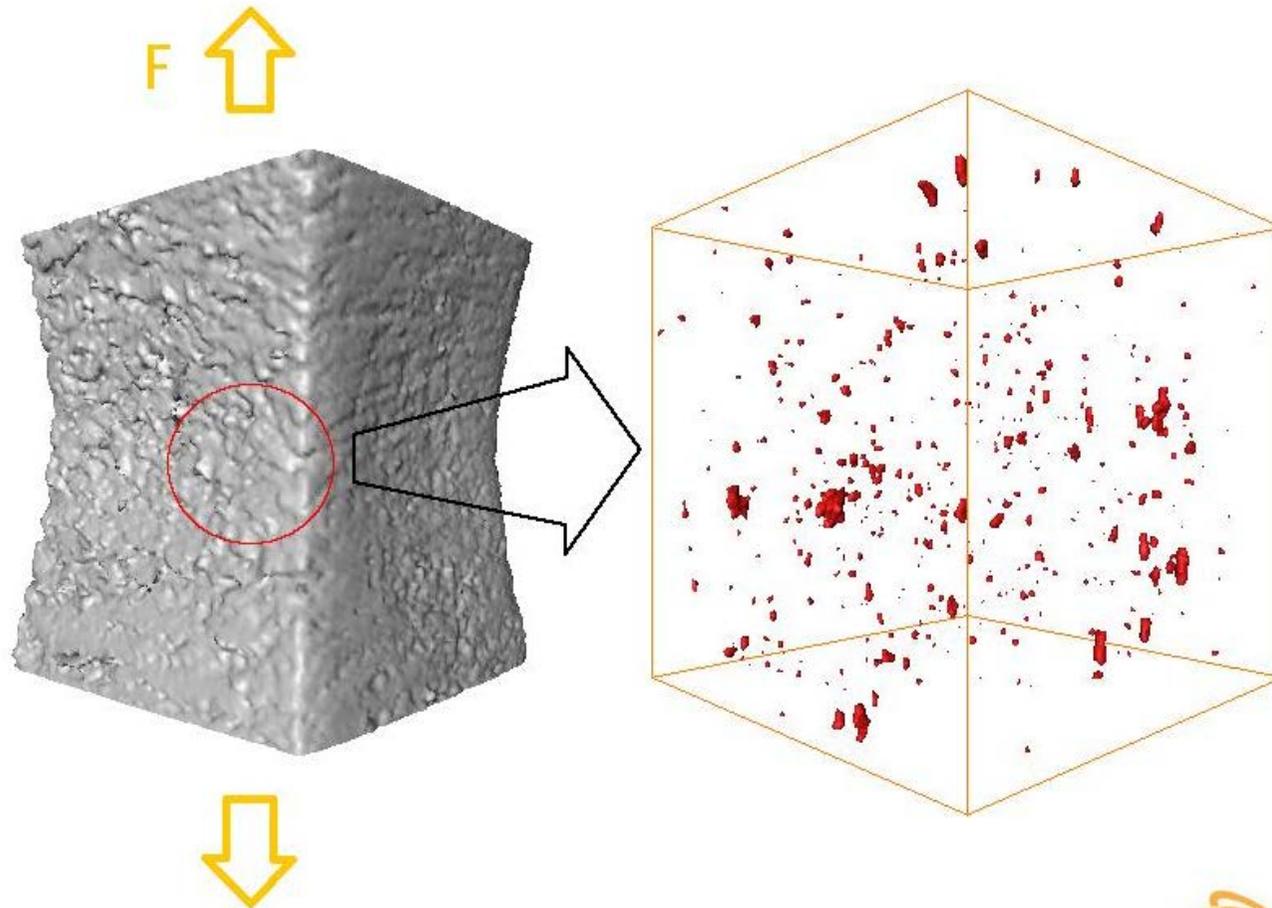
DP 600



# Typical result

1 mm sample diameter,  $\sim 1$  micron voxel size, 300 microns central box for analysis

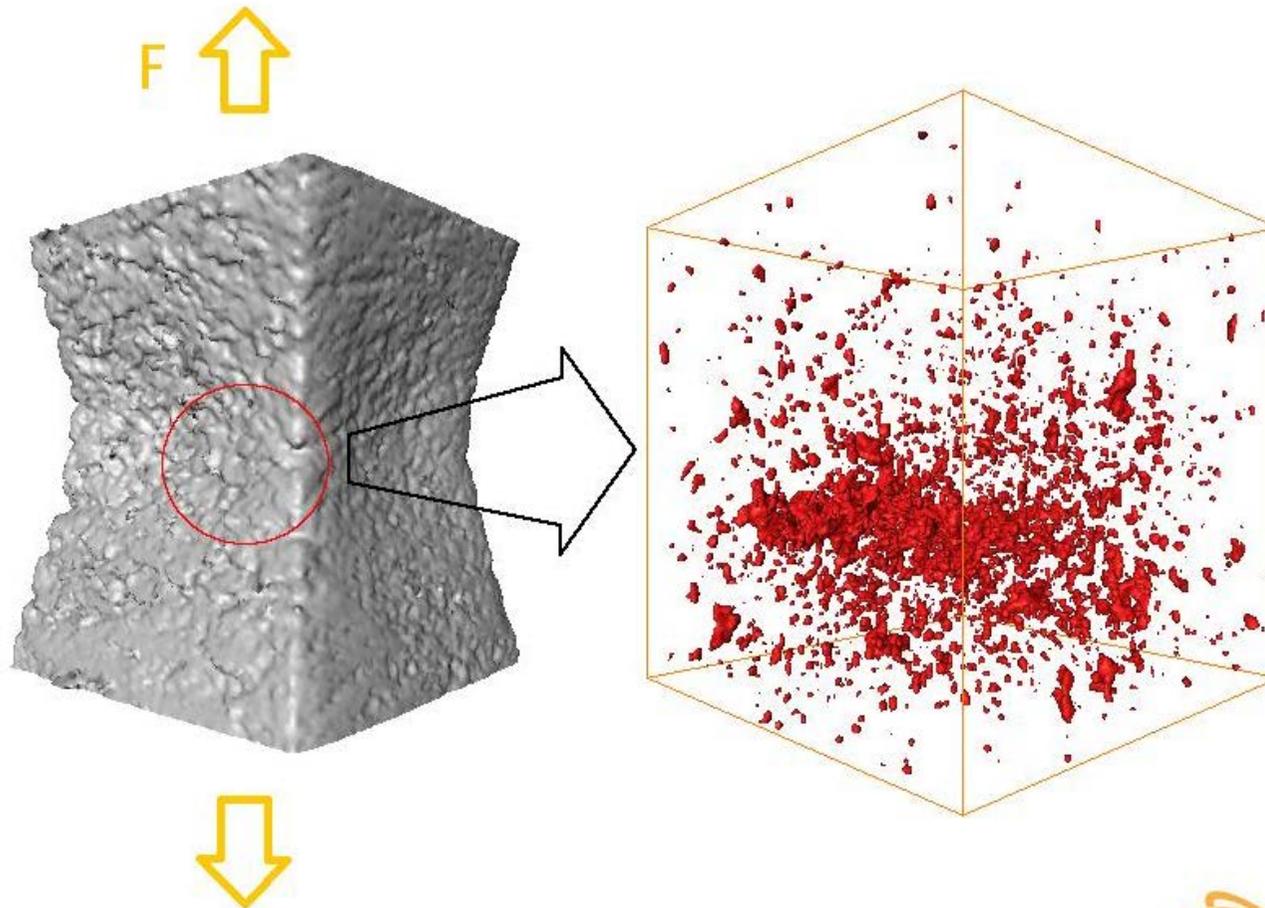
DP 600



# Typical result

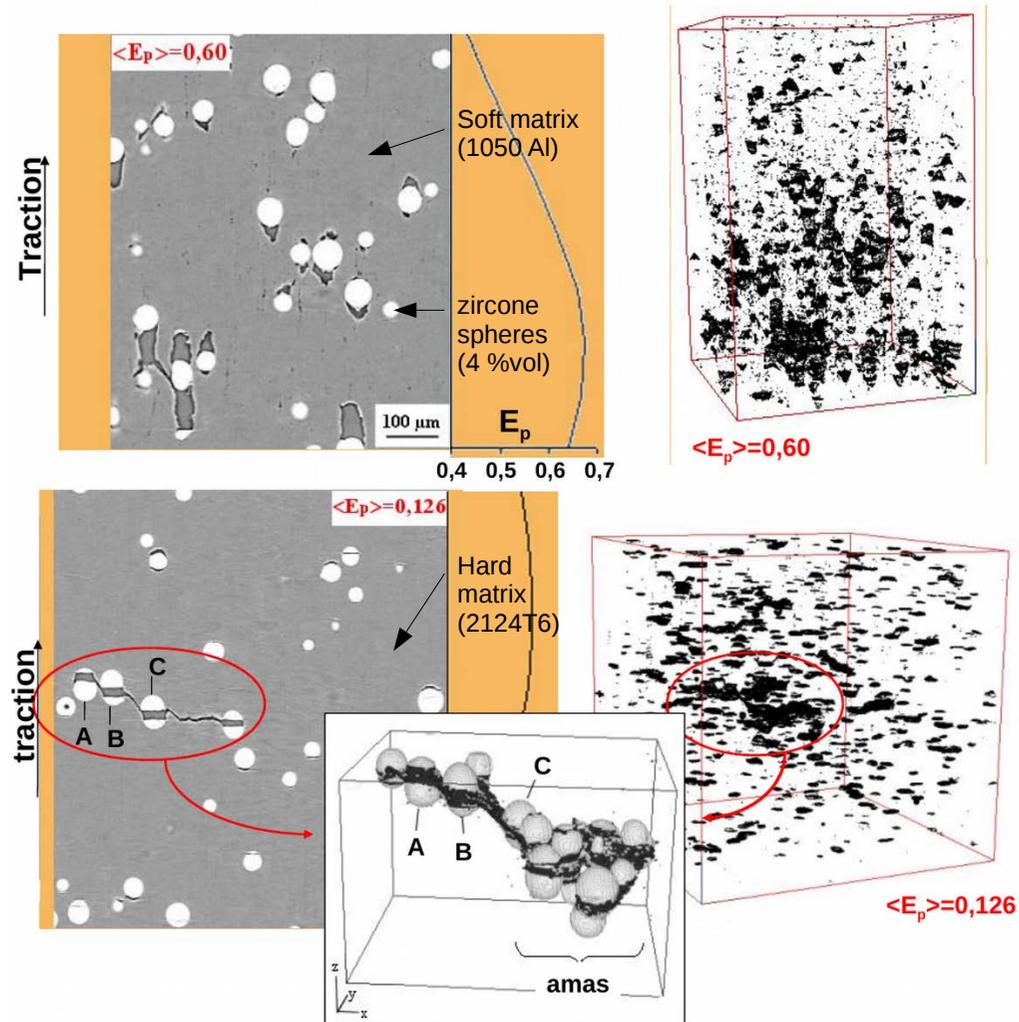
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DP 600



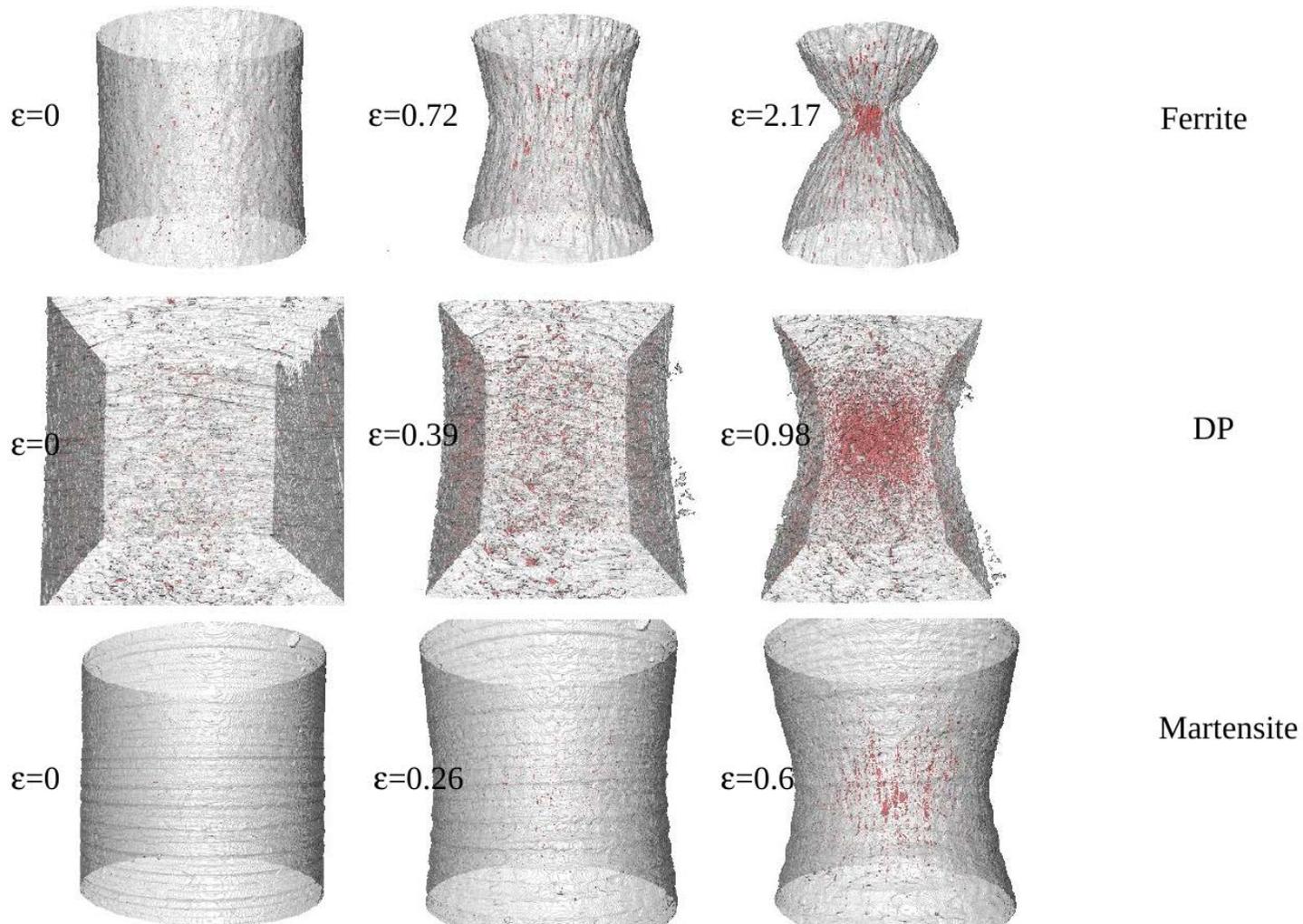
# 3D imaging: *qualitative* observations

## Model materials



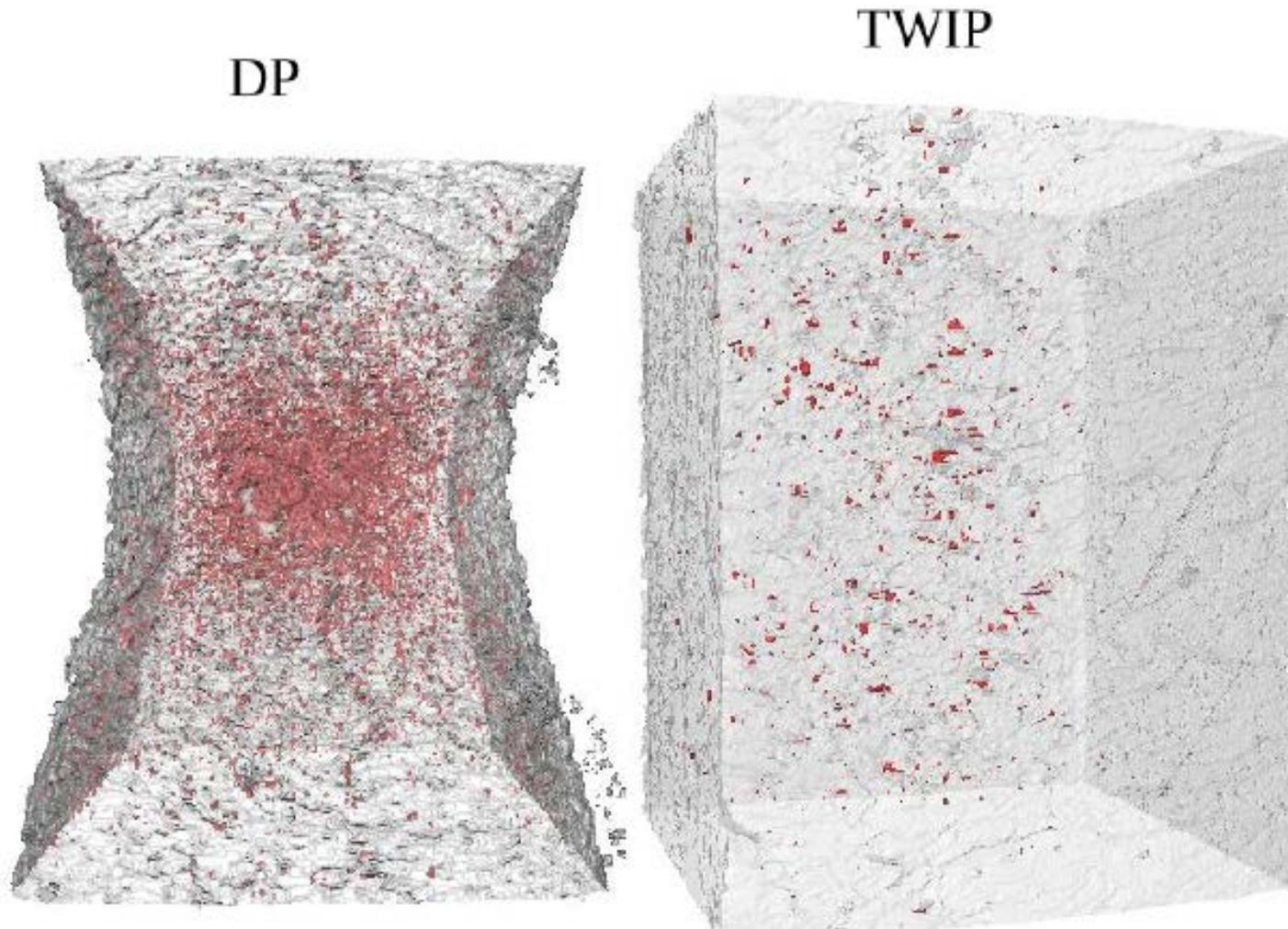
# 3D imaging: *qualitative* observations

## Ferritic steels - different hardening behaviors

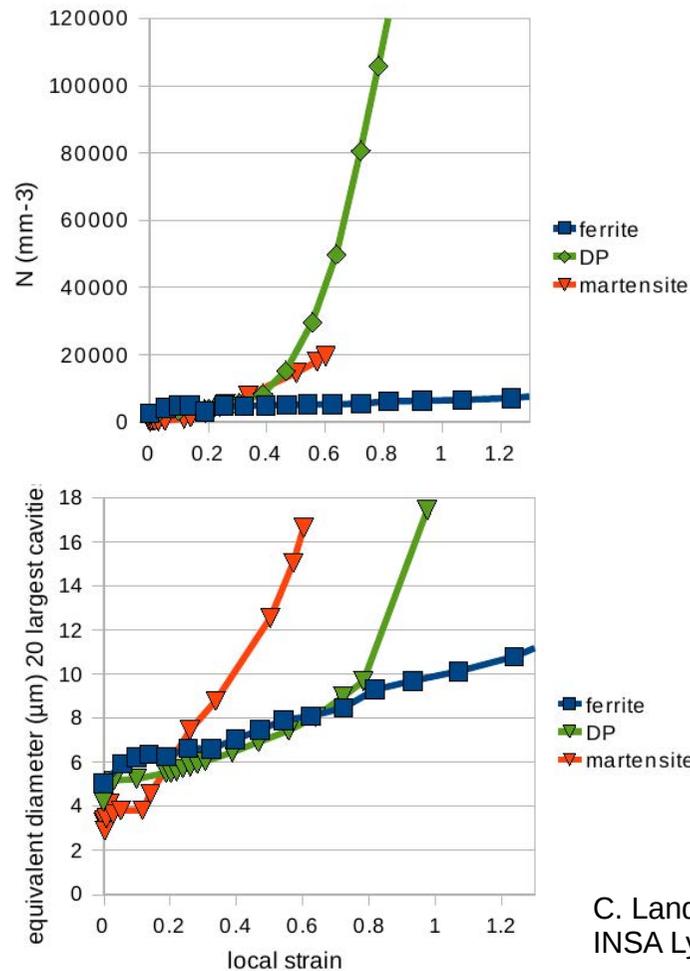


# 3D imaging: *qualitative* observations

## Different steel designs



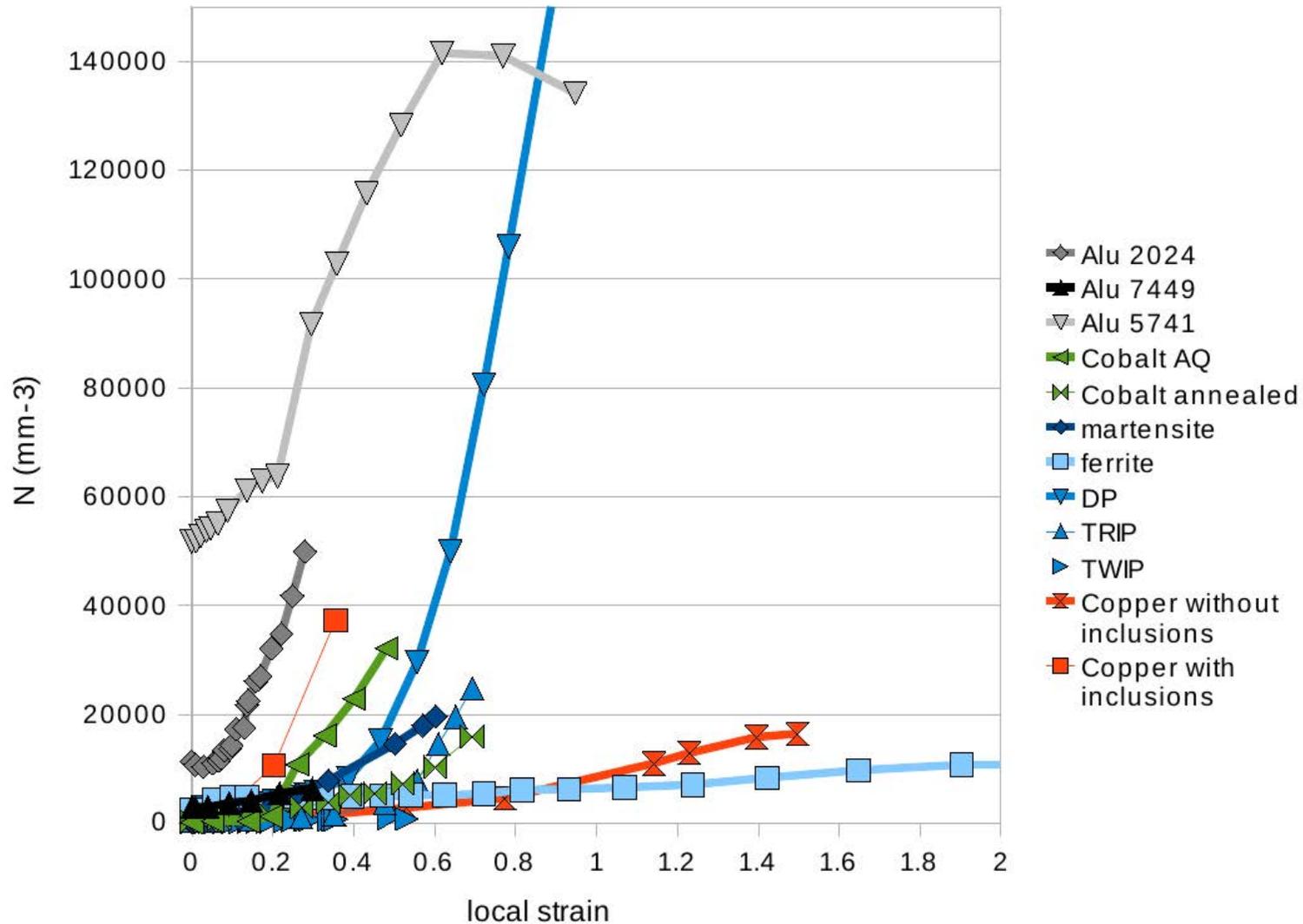
# 3D imaging: *quantitative* measurements



C. Landron PhD, INSA Lyon, 2011

- **nucleation:**  
e.g. nbr of cavities/mm<sup>3</sup>
- **growth:**  
e.g. equivalent diameter of biggest cavities.
- **coalescence:**  
e.g. cavity spacing.
- $\implies$  comparison to models!

# 3D imaging: *quantitative* measurements nucleation



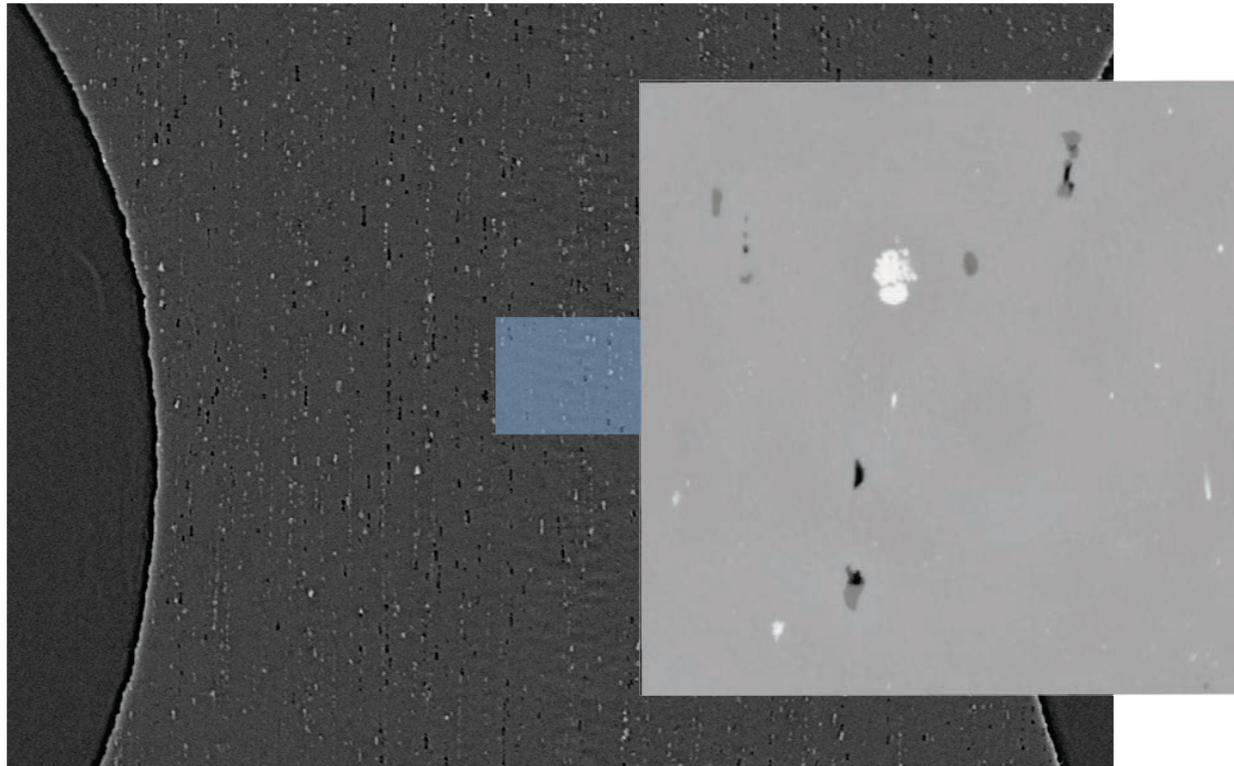


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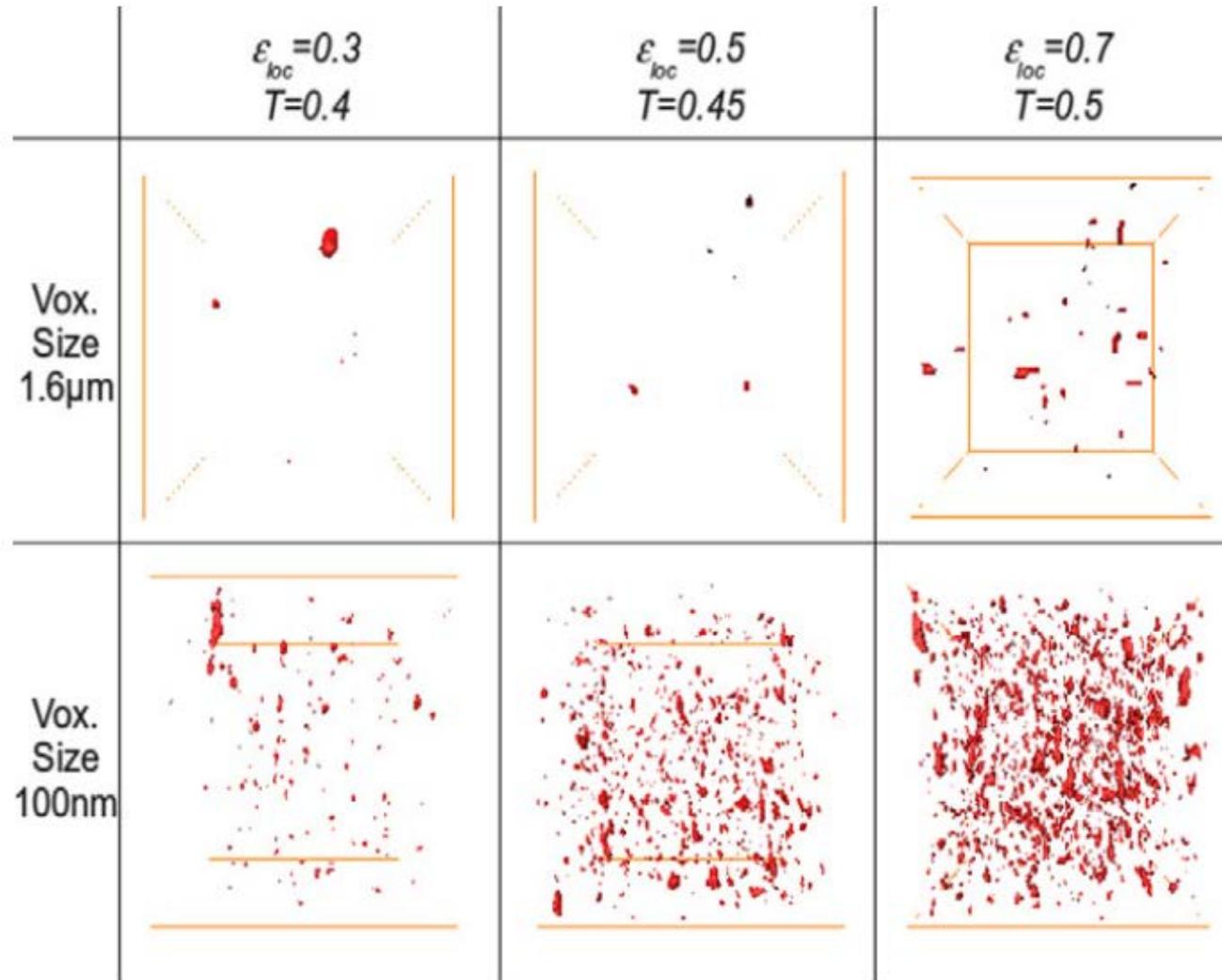
# High resolution

Down to 25 nm voxel size at ID16-B beamline, ESRF

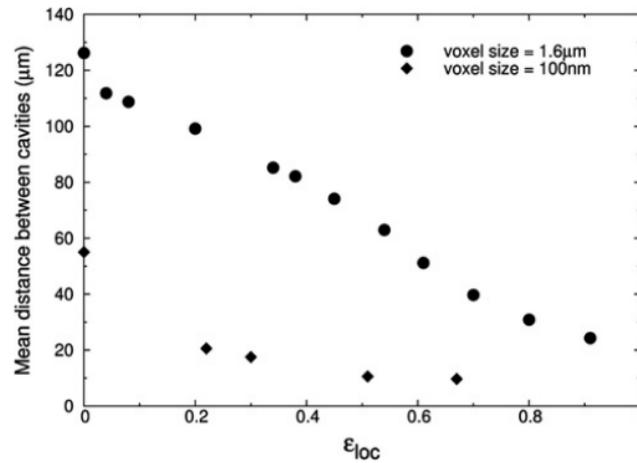
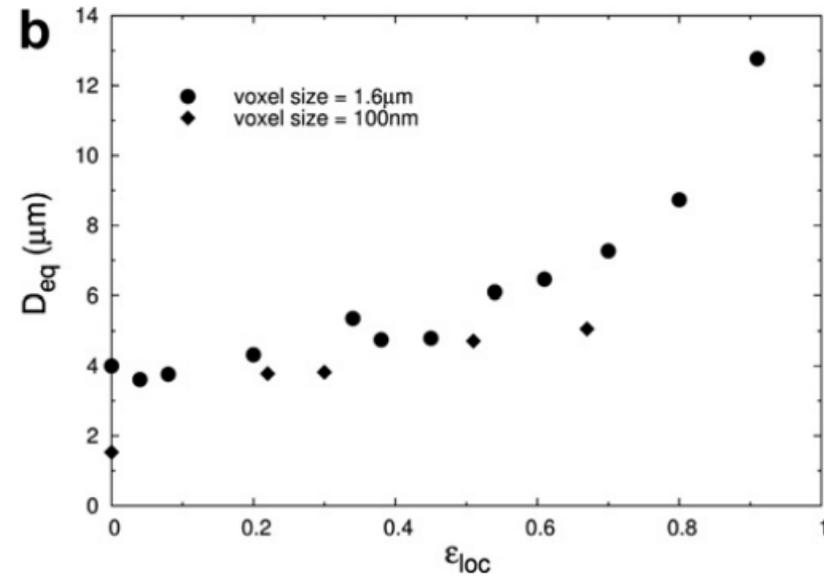
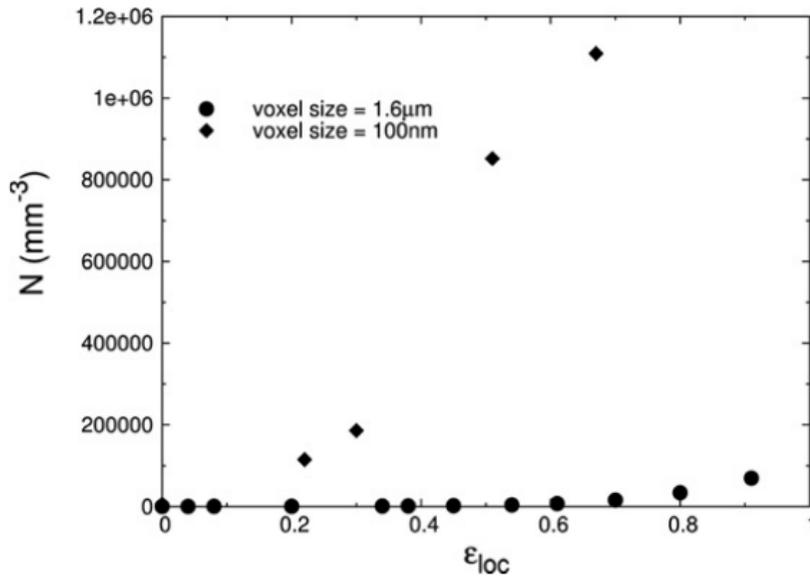


- **Local tomography:**  $\sim 50$  or  $100 \mu\text{m}$  wide region **in the sample.**
- Exploiting **phase shift** of incident X-ray beam on microstructure.

# Comparing standard and high resolution



# Quantification with high resolution



# Motivations / Open questions

## Better observing and quantifying ductile damage

- Do it in the bulk, non destructively,
- for the 3 stages of ductile damage:  
nucleation, growth and coalescence.  
⇒ attenuation and phase contrast tomography

## Effect of local microstructure

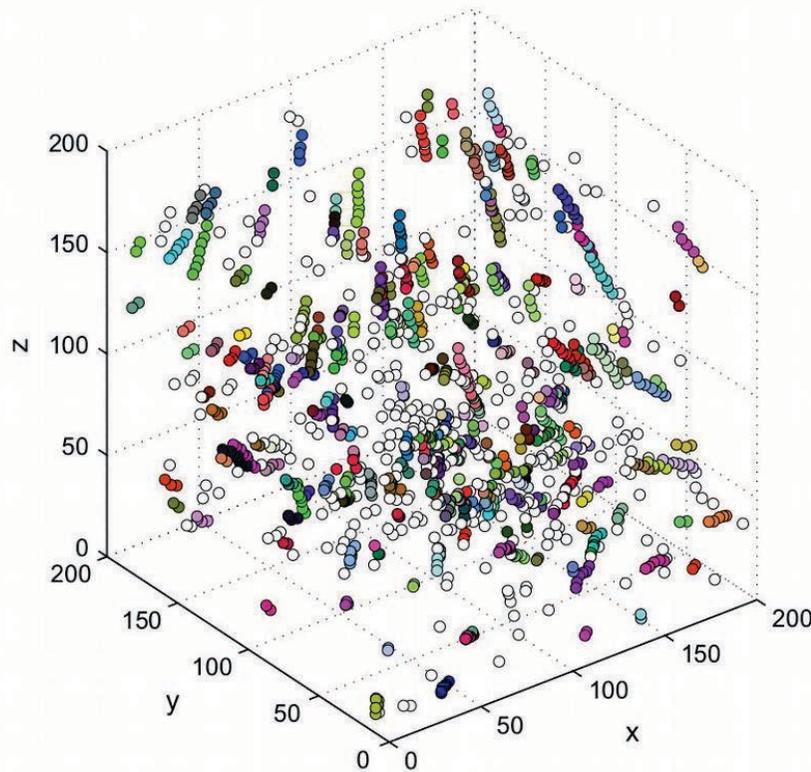
- Behavior of individual cavities during deformation ?  
How heterogeneous ?  
⇒ **automatic cavity tracking**
- Relate the individual behavior to
  - local microstructure configuration ?
  - local crystallographic orientation of grains ?

# Outline

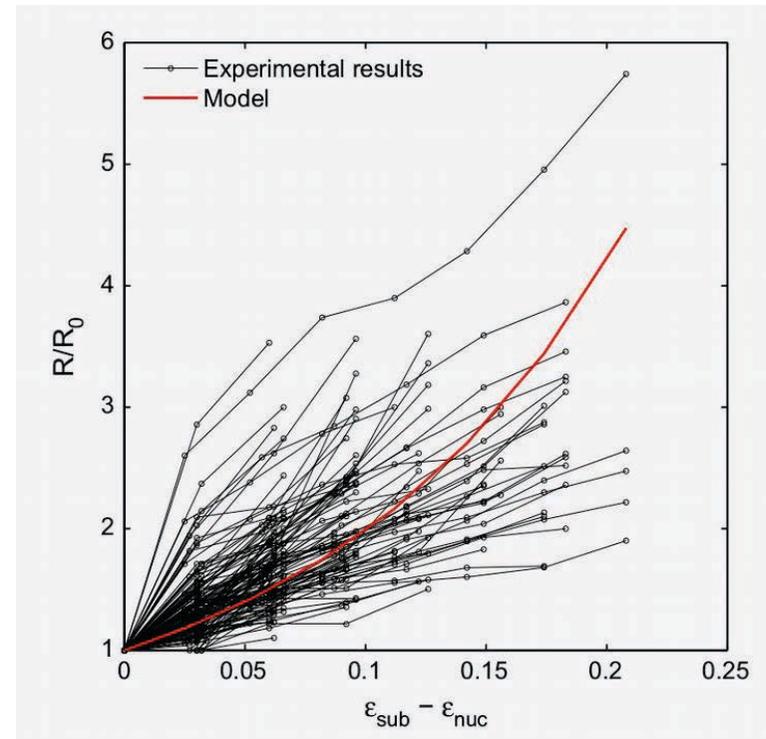
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# Cavity tracking



[Lecarme et al., Acta Mat. 2014]



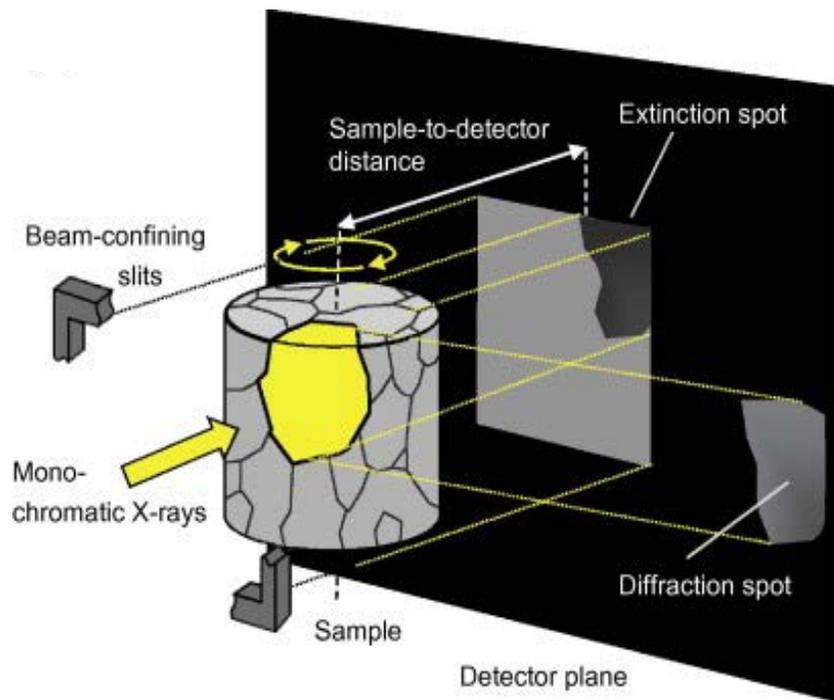
- Highly **heterogeneous** growth.
- Strong influence of local microstructure.
- Not predicted by standard damage models...



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# Diffraction Contrast Tomography (DCT)

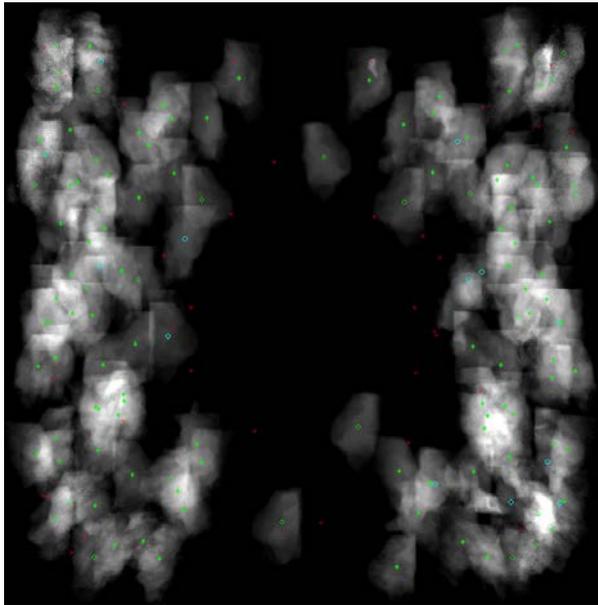


Grains on detector.

Herbig et al., Acta Mat. 2011

- Often followed by phase contrast tomography (PCT) to monitor damage development.
- Setup available at ESRF, ID11 beamline.
- [Ludwig et al., Rev. Sci. Inst. 2009]

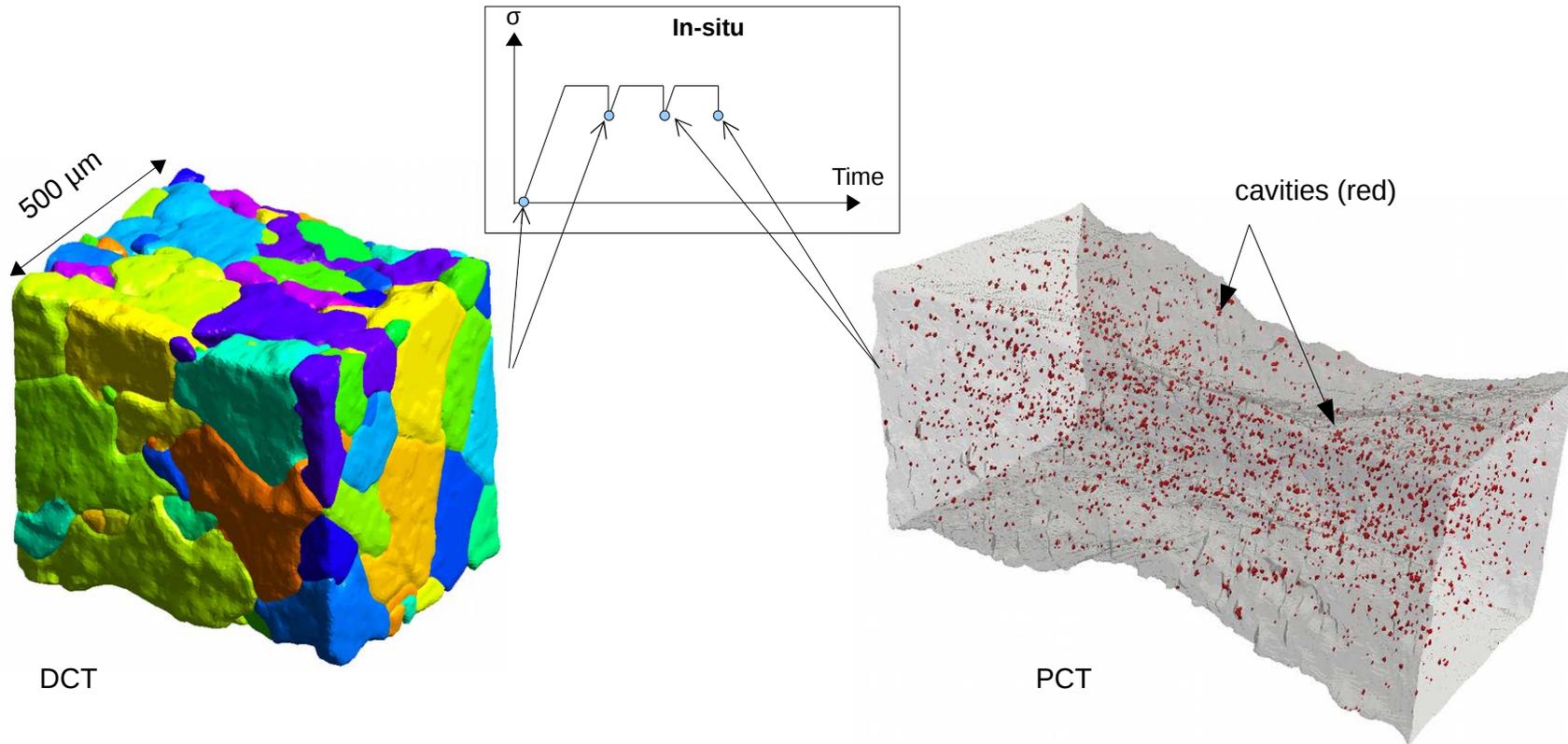
# DCT reconstruction



- Identification of  $hkl$  and  $\bar{h}\bar{k}\bar{l}$  diffraction spots (Friedel pairs) for each grain.
- Regrouping sets of pairs per grain  $\implies$  crystal orientation.
- Algebraic reconstruction of 3D grain shapes from diffraction spots.

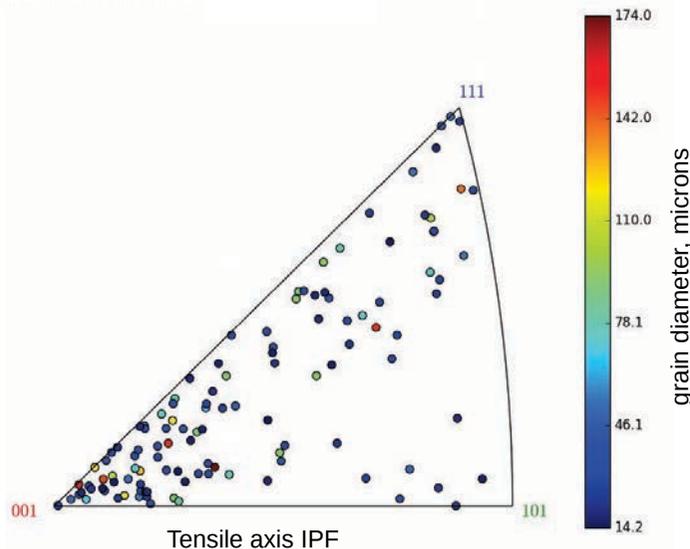
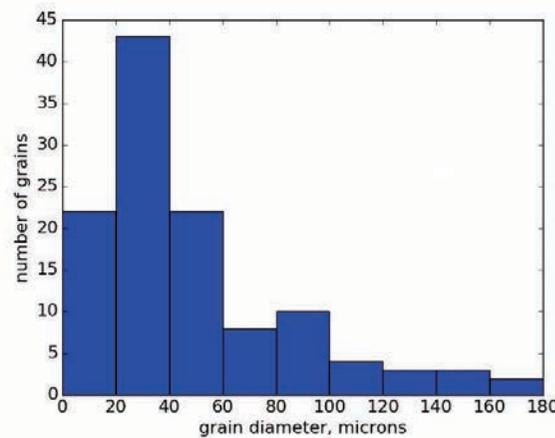
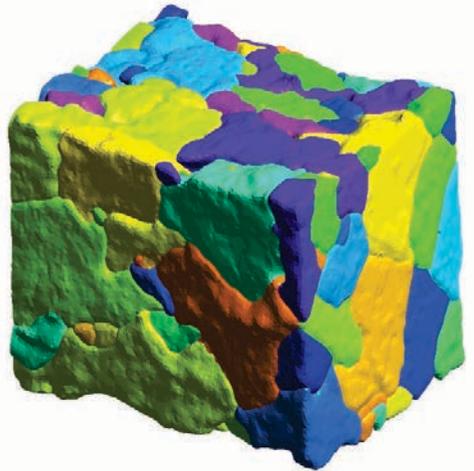


# Sample used in this illustration



- 1050 Al alloy loaded in-situ, ID11 beamline, ESRF.
- Initial state and first 2% of plastic deformation followed by DCT and far-field diffraction.
- Ductile damage further followed by PCT.

# Initial polycrystal reconstruction, DCT

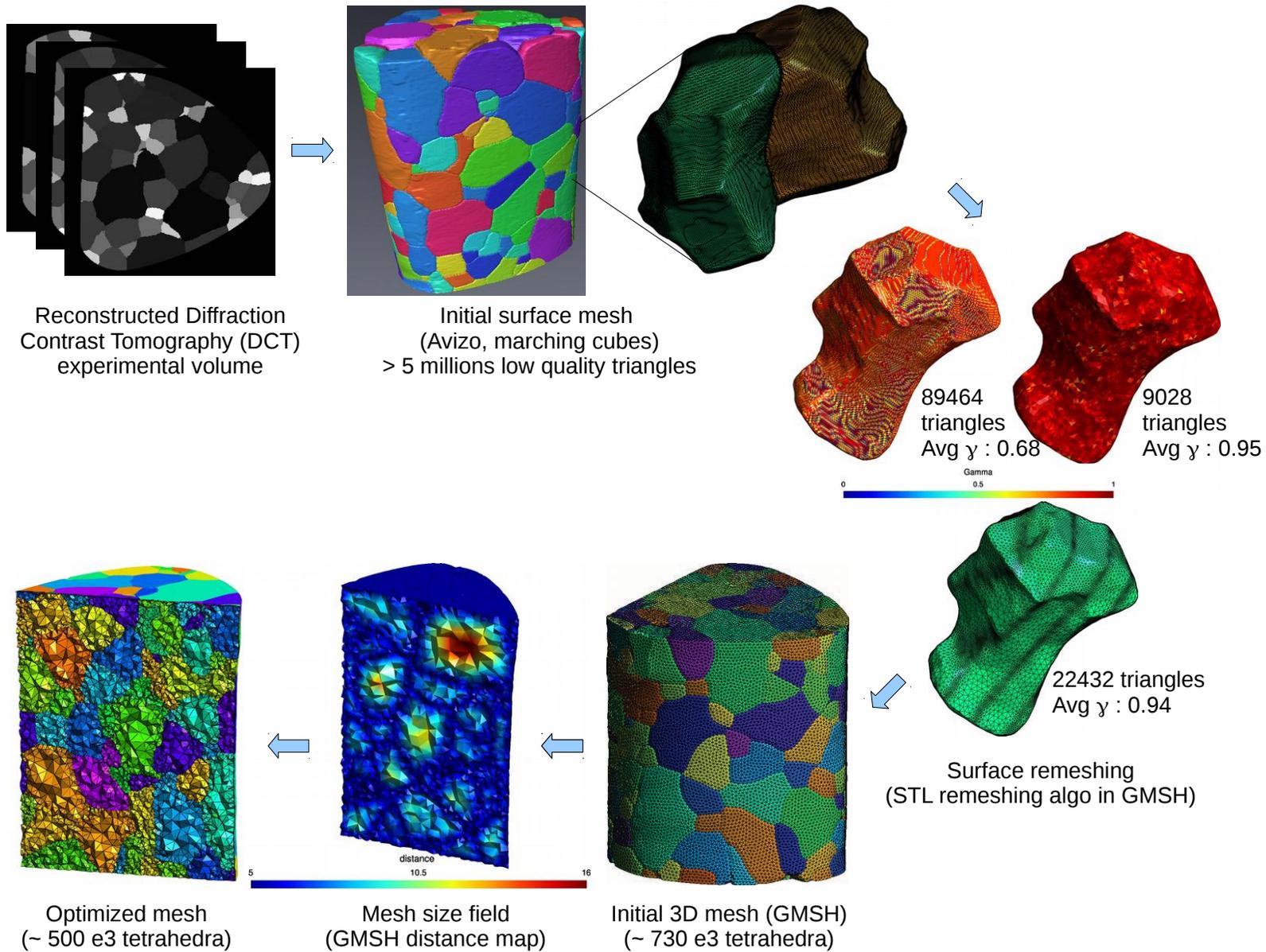


- 1050 Al alloy, recrystallized
- ~ 120 grains
- Avg diameter: 73  $\mu\text{m}$
- Average ini. disorientation:  $0.1^\circ$

# Outline

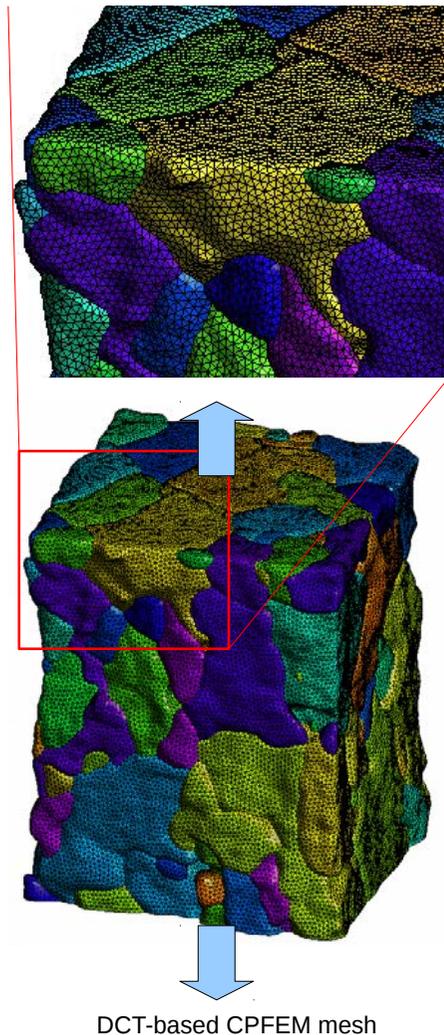
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# 3D meshing



# 1050 Al model

## Crystal Plasticity Finite Element Modeling (CPFEM)



- Elasto-viscoplastic UMAT [Delannay et al., Int. J. PLast. 2006] .
- 12 FCC  $\{111\}\langle 110\rangle$  slips systems for aluminum.
- Voce hardening of the slip systems.
- Initial grain orientation from DCT.
- Scale transition:  
DCT-based CPFEM (full field).
- 50% tensile deformation.

# CPFEM - heterogeneity of stress

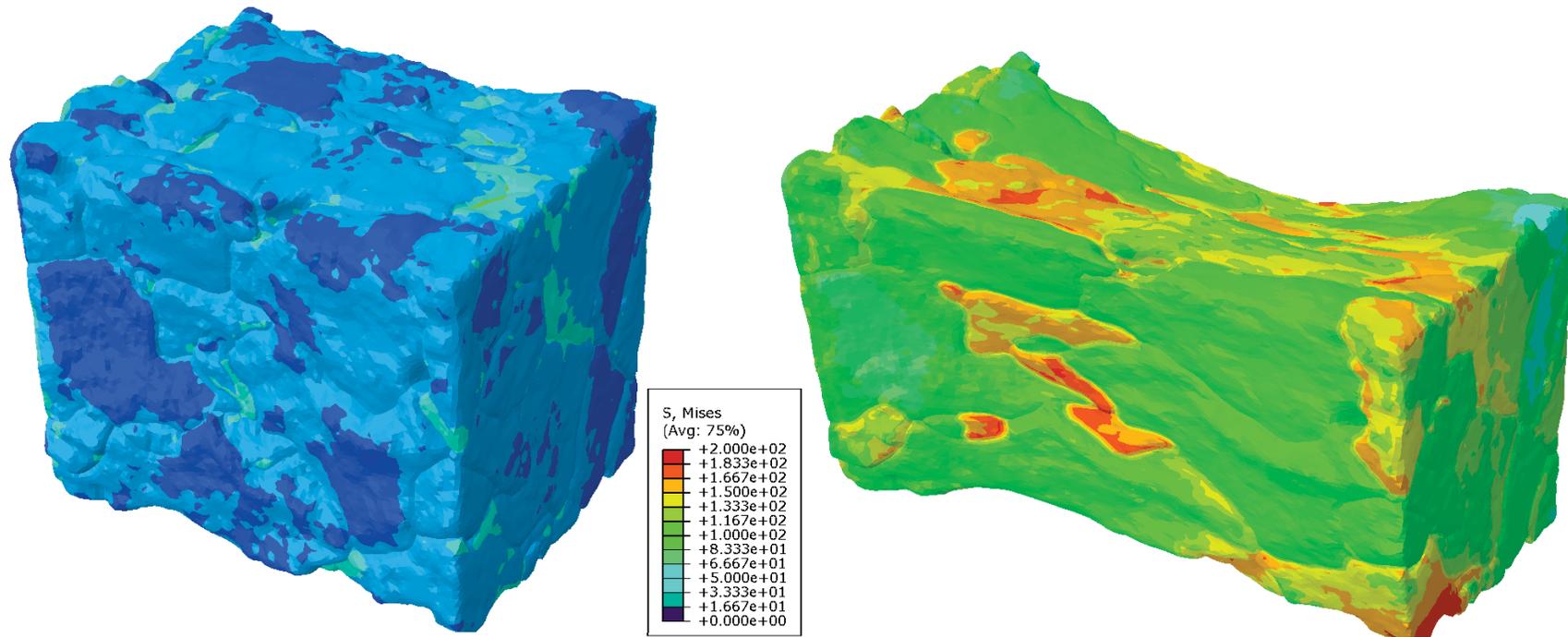
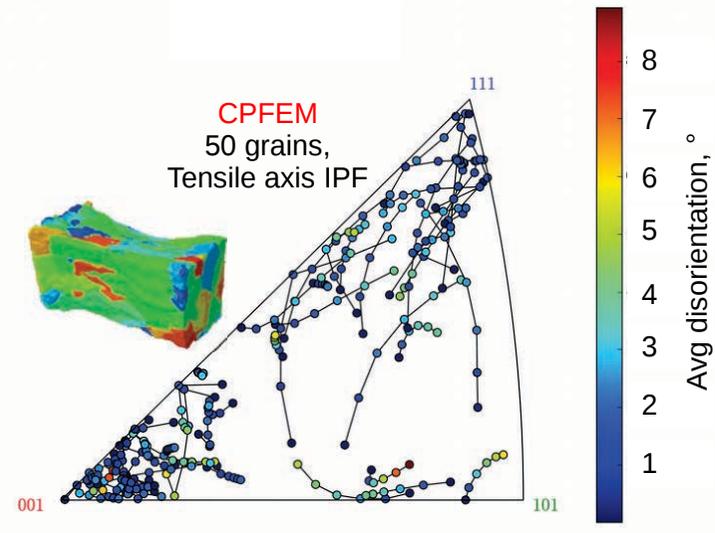
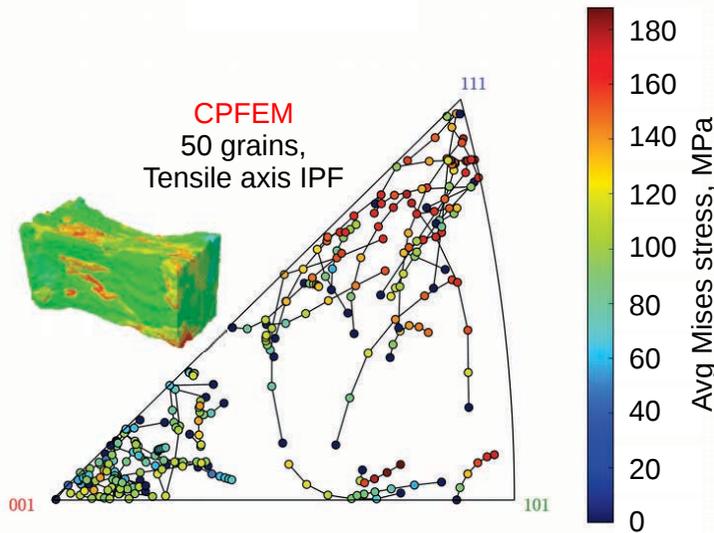
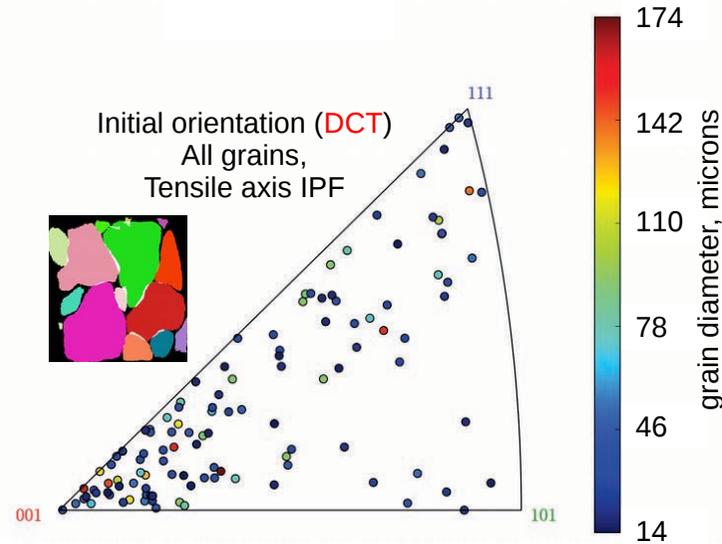


Figure: Von Mises from the initial to deformed configurations.

# CPFEM - crystal rotations





# Conclusion - Outlook

- A lot has been learned about ductile damage in the last 20 years.
- New tools are available:
  - cavity/particle tracking,
  - fast imaging,
  - high resolution,
  - but also DVC, laminography, FE-based modelling, ...
- The vision has changed.
- Still a lot to do/discover:
  - $H_2$  embrittlement,
  - new materials,
  - Development of microstructure-based models

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